Analysis of Degradation Mechanism of Lithium Iron Phosphate Battery

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Abstract

The degradation mechanisms of lithium iron phosphate battery have been analyzed with 150 day calendar capacity loss tests and 3,000 cycle capacity loss tests to identify the operation method to maximize the battery life for electric vehicles. Both test results indicated that capacity loss increased under higher temperature and SOC conditions. And also, large increase of internal resistance on the high temperature and high SOC conditions was confirmed by AC impedance tests. The real cycle capacity loss characteristic was derived by subtracting the capacity decrease due to calendar capacity loss during the cycle test from the overall capacity loss characteristic obtained from the cycle test. As a result, it is found that the real capacity loss contains not only structural disorders of electrode but also degradation factors due to the chemical reactions. Characteristics of degradation were quantified with equations based on the chemical kinetics. With this degradation prediction, an operation method was proposed that is compatible with the long life of batteries and the safety driving of a vehicle. As a result, with optimizing the SOC range used in the operation as follows: 30-10% in the warm seasons, 45-25% in the cold seasons, it was found that batteries can last 4 times longer than it used with high SOC range in every season.

Keywords: lithium battery, battery calendar life, BEV (battery electric vehicle)

1 Introduction

In recent years, battery electric vehicles (BEVs) have been actively developed and introduced to the market. BEVs have superior environmental performance because emit no CO\textsubscript{2} and other gases while driving. However, in terms of more popularization of BEVs, weaknesses for cruising range, cost of batteries and user-friendliness of charging remain major challenges. To resolve these issues, a concept “short range driving and very frequent charging” was formulated [1]. Under this approach, the authors developed the short-range frequent-recharging electric vehicles (Figure 1) and studied methods of greatly reducing the number of large, heavy, and expensive batteries. In addition, researching about non-contact inductive power supply (IPS) system was developed as a way of improving charging convenience, and a lithium iron phosphate battery was developed suit to rapid charging [2]. Especially about batteries, a lot of efforts have been put into analyzing its degradation mechanisms. Up to now, our research group has been made various studies about batteries for BEVs [3]. By analyzing the degradation mechanism of batteries, it could be possible to
obtain guiding principles for next generation batteries and indicate how to last the life of batteries. Also, battery degradation causes problems such as decline of cruising range and decrease of power. In the scope of safety operation, it is significant that controlling the progressing of battery degradation. The aim of this paper is to contribute the longevity of BEVs with addressing the operation method to realize the coexistence vehicle’s safety operation and battery’s long life.

2 Evaluation Tests

2.1 Degradation Factors
Degradation factors of lithium-ion batteries have been reported by previous researches [4] [5]. Figure 2 shows the waste of lithium-ion comes from formation of solid electrolyte interface (SEI). Formation of SEI is the dominant degradation factor in the calendar capacity loss. Figure 3 shows the structural disorder of the electrode as lithium-ion goes in and out of the electrode. Structural disorder is progressing with the discharge and charge processes. Furthermore, acceleration of wasting lithium-ion with the discharge and charge processes could be the factor of degradation in the cycle processes. Temperatures, state of charge (SOC) and SOC range in the tests were extracted as external factors that have impacts on degradation.

\[
\text{Li}^+ + \text{electrolyte} \rightarrow \text{SEI}
\]

\[
\begin{align*}
\text{ROCO}_2\text{R} + \text{Li}^+ + e^- & \rightarrow \text{ROCO}_2\text{Li} + \text{R}^+ \\
\text{ROCO}_2\text{Li} + \text{Li}^+ + e^- & \rightarrow \text{Li}_2\text{CO}_3 + \text{R}^+ \\
\text{LiPF}_6 + 3\text{ROCO}_2\text{Li} + \text{H}_2\text{O} & \rightarrow 2\text{HF} + \text{R}_3\text{PO}_4 + 3\text{CO}_2 + 4\text{LiF}
\end{align*}
\]

Figure 2: SEI Reaction Model

Figure 3: Structural Disorder Model

2.2 Calendar Capacity Loss Tests

Table 1: Test Battery Specification

<table>
<thead>
<tr>
<th>Cathode Material</th>
<th>LiFePO₄</th>
</tr>
</thead>
<tbody>
<tr>
<td>Anode Material</td>
<td>C₆ (Graphite)</td>
</tr>
<tr>
<td>Rated Voltage</td>
<td>3.25 V</td>
</tr>
<tr>
<td>Rated Capacity</td>
<td>6.2 Ah</td>
</tr>
<tr>
<td>Dimensions (mm)</td>
<td>L120 × W3 × H140</td>
</tr>
</tbody>
</table>

Figure 4 and Table 1 show the specification of the test battery cell. Lithium iron phosphate battery is known for its superiority of safety and manufacturing cost. Test battery cell is laminate
type cell, has superior rapid charging performance and is suitable for electric vehicles designed to be charged frequently and driven short distances between charges. Parameters for the calendar capacity loss tests are temperature (5°C, 25°C, 45°C) and SOC (90%, 50%, 10%). Figure 5 shows the results of the calendar capacity loss tests. These results indicated that the capacity loss increased under the higher temperature and SOC conditions. It is thought that the main cause of this large capacity loss is that the reactions between the lithium ion and the electrolyte are accelerated under high temperature and SOC conditions. As a result, the absolute amount of lithium ions decreases due to the growth of SEI films.

![Figure 5: Calendar Capacity Loss Test Results](image)

(a) Temperature Dependency

(b) SOC Dependency

Figure 5: Calendar Capacity Loss Test Results

### 2.3 Cycle Capacity Loss Tests

Table 2 shows the condition of cycle capacity loss tests. For the cycle tests, the cell was discharged from each start SOC and charged to the start SOC. The charge rate is 2.4C (14.88A) and the discharge rate is 1.2C (7.44A). Also after charging, 5 minutes rest has set to avoid sudden temperature rise. Charge/discharge amount is 1.24Ah that equivalent to 20% SOC of test cell. Charge/Discharge rate and Amount Ah of a cycle are determined by assuming the operation way of WEV-1. The cut-off voltage for the cycle tests and capacity measurement tests were at 4.0V and 2.0V. Capacity measurement test was carried out on the condition of 25°C. Parameters for the cycle tests are temperature (5°C, 25°C, 45°C) and SOC range (90-70%, 70-50%, 40-20%). Test battery cell is same type one used in the calendar capacity loss tests, expect has about 12% of initial capacity loss in the cause of 900 day, 100 cycle running tests.

![Figure 6: Cycle Capacity Loss Test Results](image)

(a) Temperature Dependency

(b) SOC Dependency

Figure 6: Cycle Capacity Loss Test Results

<table>
<thead>
<tr>
<th>Temperature</th>
<th>Temperature</th>
<th>SOC range</th>
<th>SOC range</th>
</tr>
</thead>
<tbody>
<tr>
<td>5°C, 25°C, 45°C</td>
<td>5°C, 25°C, 45°C</td>
<td>90-70%</td>
<td>90-70%</td>
</tr>
<tr>
<td></td>
<td></td>
<td>70-50%</td>
<td>70-50%</td>
</tr>
<tr>
<td></td>
<td></td>
<td>40-20%</td>
<td>40-20%</td>
</tr>
</tbody>
</table>

<table>
<thead>
<tr>
<th>Charge/Discharge Rate</th>
<th>14.88A/7.44A</th>
</tr>
</thead>
<tbody>
<tr>
<td>Charge/Discharge Amount</td>
<td>1.24Ah/1.24Ah</td>
</tr>
</tbody>
</table>
2.4 AC Impedance Tests

AC impedance tests were carried out to analyze the degradation of test battery cell. AC impedance test is one of the transient measurement methods that impedance and admittance could be calculated by measuring the response to the micro AC signal applied to the battery. An equivalent circuit shown in Figure 7 was used for analyzing the impedance spectrums. $L$ is inductor factor of measurement systems, $R_{sol}$ is resistance of electrolyte, $R_{ct}$ is resistance of moving electric charge, $Z_{w}$ is Warburg impedance of diffusion, $CPE$ is kind of an electric double-layer capacity that expresses elliptic distortion of plots [6]. This circuit was selected to fit the measured impedance spectrum.

Figure 7: Equivalent Circuit of Test Battery Cell

All tests were carried out on the condition of discharge at 50% SOC, frequency range between 0.1 and 10k Hz. Figure 8 shows the cole-cole plots of test batteries. These plots contain both cathode and anode reactions. Shifting of the colec plots along the axis $Z'$ indicates increase of the $R_{sol}$, and also expand of radius of the semicircle indicates increase of the $R_{ct}$. Increase of the $R_{sol}$ has shown significantly under high temperature and SOC conditions. This is speculated to cause of grown SEI films and could be the main factor of degradation. And also a little increase of the $R_{ct}$ has shown under high temperature and SOC conditions. This is speculated to cause of structural disorder of electrode. Increase of internal resistance attend to capacity fade has confirmed by evaluating absolute amount of $R_{sol}$ and $R_{ct}$.

Figure 8: Cole-Cole Plots of Test Battery Cells

3 Discussion

3.1 Calendar Capacity Loss

Figure 5 suggests that chemical reactions are the main cause of calendar capacity loss. Therefore, characteristics of calendar capacity loss were quantified with the equation based on the chemical kinetics. Previous research has suggested that capacity loss is relevant to the amount of grown SEI films. A simple model for generation of SEI films is represented by the equation (1). The reaction rate of equation (1) moves from equilibrium state is quantified with the equation (2).

$$\text{Li} + \text{electrolyte} + e^- \rightarrow \text{SEI} \quad (1)$$

$$v_0 = A'[Li^+]e^{-E_a\over RT} \quad (2)$$

Where $A'$ is frequency factor, $E_a$ is activation energy, $R$ is gas constant, and $T$ is ambient temperature.

While terminal voltage rises by $\Delta V$ from equilibrium state, the electric potential of anode drops by $\Delta E$ and potential energy of electron on the surface of anode drops by $F\Delta E$. This variation of potential energy becomes the motive force of reduction reaction which generates SEI films. Activation energy drops by $(1 - \beta)F\Delta E$. Reaction rate equation while the terminal voltage rises by $\Delta V$ is quantified with the equation (3).
\[ v = A' [\text{Li}^+] e^{-\frac{E_a - (1-\beta)\Delta F AE}{RT}} \]  

\( \Delta V = V \) could be completed where reference voltage \( V_0 \) is defined as 0V. To simplify the equation, here suppose that \( \Delta E \propto \Delta V = V \). Equation (3) could be expressed as equation (4) if the consistency of Li-ion is proportional to the SOC.

\[ v = A(SOC) e^{-\frac{E_a - \alpha V}{RT}} \]  

Each coefficients \( A, E_a, \alpha \) are determined with curve fitting techniques. Characteristics of calendar capacity loss were quantified with equation (5).

\[ k_s = 4475 \times (SOC) \times \exp\left(\frac{-49767 - 811V}{RT}\right) \]  

The variable \( k_s \) in the equation is defined as the calendar capacity loss coefficient.

### 3.2 Cycle Capacity Loss

The calendar capacity loss progresses even at the cycle capacity loss tests. The real cycle capacity loss characteristic was derived by subtracting the capacity decrease due to calendar capacity loss during the cycle test from the overall capacity loss characteristic obtained from the cycle test. Figure 7 shows that the real capacity loss progresses linearly with the square root of the number of days of cycles. From the temperature and SOC dependency shown in Figure 7, it is suggested that the real capacity loss contains not only structural disorder but also a factor due to the chemical reactions. Separation and regeneration of SEI films progressing with the charge and discharge processes are speculated to be the cause of this chemical reaction. Characteristics of cycle capacity loss were also quantified with the equation (6) based on the mechanical disorder and chemical kinetics.

\[ k_c = 394.1 \times (SOC) \times \exp\left(\frac{-31013 - 0.01734V}{RT}\right) + k_{c, \text{mechanical}} \]  

Where \( N \) is cycle number, \( k_c \) is defined as the cycle capacity loss coefficient. \( k_{c, \text{mechanical}} \) is fixed number shown in equation (7).

\[ k_{c, \text{mechanical}} = 0.154\ (5^\circ C), 0.0855\ (25^\circ C,\ 45^\circ C) \]

Figure 7: Real Capacity Loss in Cycle Tests

### 3.3 Optimization of the BEV’s Operation Method

The results of calendar tests and cycle tests showed that it is desirable using in low SOC range to reduce the capacity decrease under every temperature conditions. On the other hand, an operation with low SOC range has a risk of reaching the lower limit of supply voltage especially at low temperatures because of increasing of the internal resistance. The operation of the WEV-1 was optimized with degradation.
predictions. WEV-1 was developed as a transfer tool in a vast tract of site such as premises of university or factory. Its assumed operation way is 5 round trips a day with a route consuming electric power equivalent to 20% of SOC. In this operation way, the SOC range used in the operation was optimized by seasons as follows: 30-10% in the warm seasons, 45-25% in the cold seasons. Considering the increasing of internal resistance of batteries, used SOC range in the cold seasons was set to be higher than it in the warm seasons. The life of battery was set to 70% capacity retention of fresh battery. As a result, it was found that batteries can last 4 times longer than it used with high SOC range in every season (Figure 8).

4 Conclusion

Degradation mechanisms of lithium iron phosphate battery were analyzed and an operation method of BEV was optimized with the degradation prediction. The obtained results are as follows:

(a) 150 day calendar capacity loss tests have been carried out. Experimental results showed that calendar capacity loss increased under the higher temperature and SOC conditions, and progresses linearly with the square root of time. From the results, it could be said that calendar capacity loss is caused by chemical reactions. Formation of SEI films is thought to be the dominant factor. Calendar capacity loss characteristics were quantified with an equation based on the chemical kinetics.

(b) 3,000 cycle capacity loss tests have been carried out. Experimental results showed that calendar capacity loss progresses linearly with the square root of cycle numbers. From the results, it could be said that cycle capacity loss includes both degradation due to the chemical reaction and structural disorder.

(c) The real cycle capacity loss characteristic was derived by subtracting the capacity decrease due to calendar capacity loss during the cycle test from the overall capacity loss characteristic obtained from the cycle test. As a result, the real capacity loss also progressed linearly with the square root of cycles. From the results, some chemical reactions were thought to be included in the real cycle capacity loss. Separation and regeneration of SEI films progressing with the charge and discharge processes are speculated to be the cause of this chemical reaction. The real cycle capacity loss characteristics were quantified with an equation with the mechanical disorder and chemical kinetics.

(d) With degradation predictions, an operation method was proposed that is compatible with the long life of batteries and the safety driving of a vehicle. As a result, with optimizing the SOC range used in the operation as follows: 30-10% in the warm seasons, 45-25% in the cold seasons, it was found that batteries can last 4 times longer than it used with high SOC range in every season.

In addition, large capacity loss was seen in the cycle capacity loss tests on the condition of low temperature and low SOC range. A detailed analysis on structural disorder characteristic under low temperature condition will be presented in a future paper.

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References


vehicle equipped with non-contact inductive power supply system and LiFePO4 lithium-ion battery”, Proceedings of the IEEE VPPC 2010, conference DI-1-6, pp. 1-3 (CD-ROM) (2010)


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